

High-resolution IXS to resolve puzzles of glasses

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European Synchrotron Radiation Facility



Content:

- 1. The development of high-resolution IXS at the ESRF (ID18):
monochromators and methods.**
- 2. An example of application of the high-resolution IXS
to glass dynamics.**
- 3. Scientific cases in glass dynamics for high-resolution IXS.**
- 4. Summary**

Nuclear Resonance Scattering:

(resonant scattering of x rays by nuclei with low energy (10-100 keV) nuclear transitions)

It requires ~meV monochromatization for:

- (i) measurements of time spectra of nuclear scattering (to decrease load on detector)
- (ii) measurements of density of vibrational states (for good energy resolution)

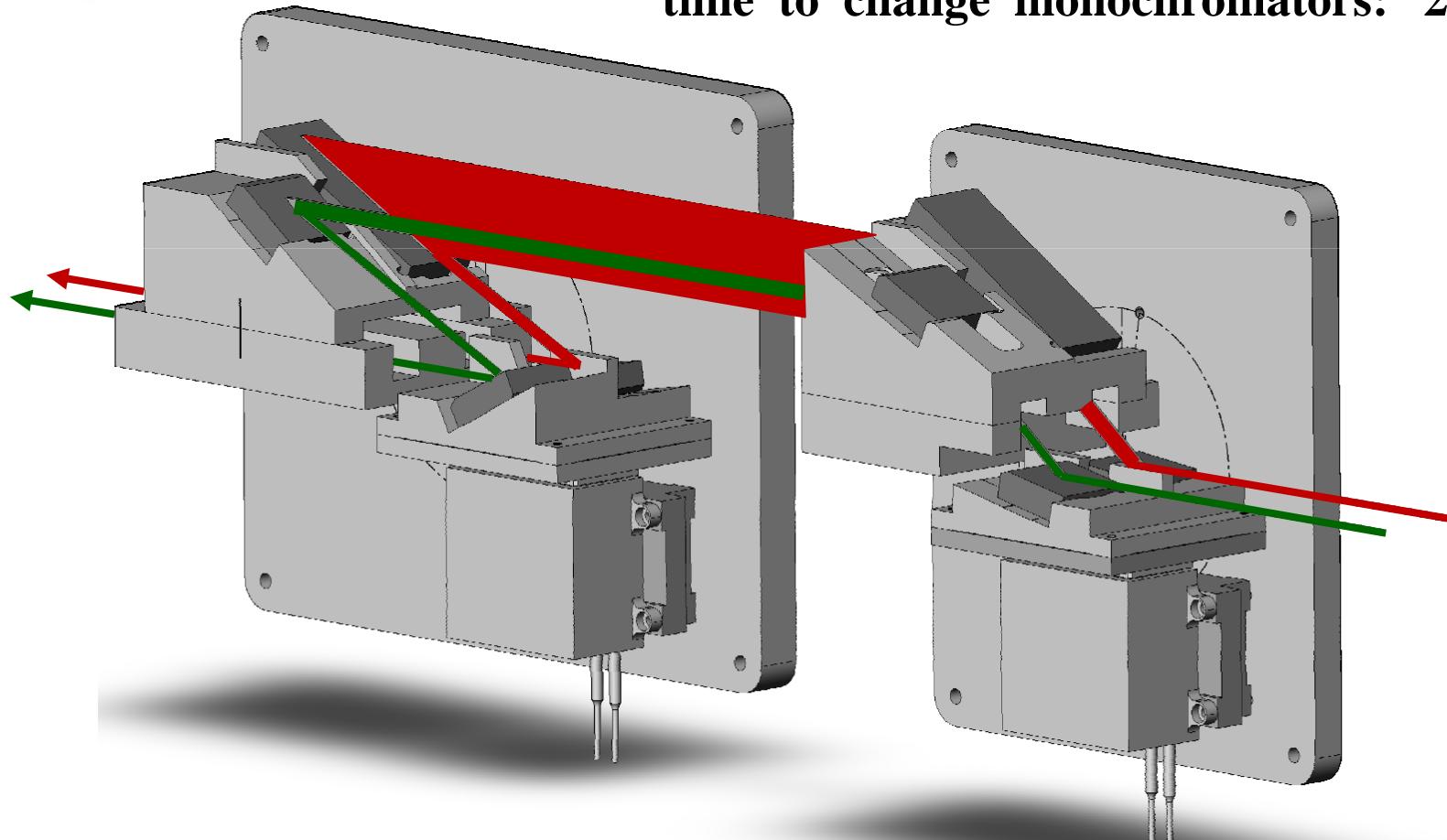
Energy [keV]	Isotope	Resolution [meV]	design
14.413	⁵⁷ Fe	6.4	nested
14.413	⁵⁷ Fe	3.1	nested
14.413	⁵⁷ Fe	1.8	in-line
14.413	⁵⁷ Fe	0.5	in-line
22.502	¹⁴⁹ Sm	0.76	nested
21.542	¹⁵¹ Eu	1.6	nested
23.879	¹¹⁹ Sn	0.65	nested
25.651	¹⁶¹ Dy	0.9	nested
27.890	¹²⁹ I	1.0	nested
35.493	¹²⁵ Te	1.1	backscattering
37.1298	¹²¹ Sb	1.2	backscattering
39.5821	¹²⁹ Xe	1.4	backscattering
67.41	⁶¹ Ni	120	nested
68.752	⁷³ Ge	24	nested

Double In-line monochromator for $E = 14.413 \text{ keV}$:

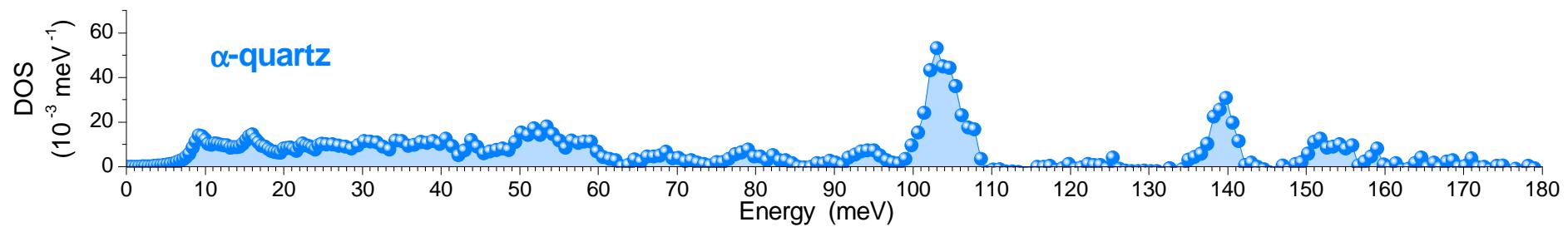
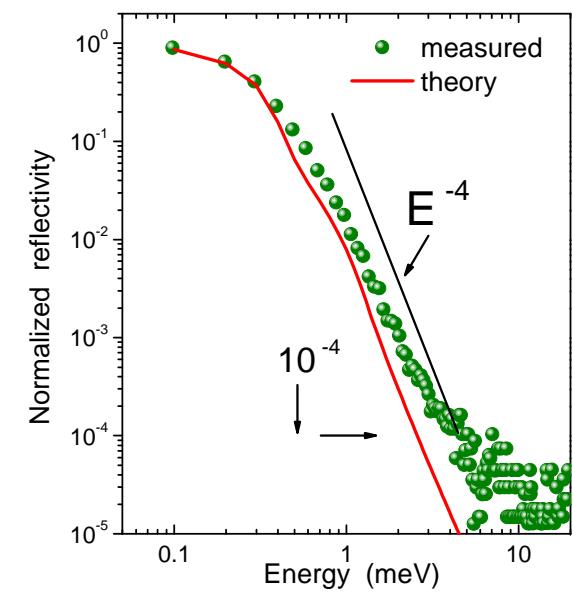
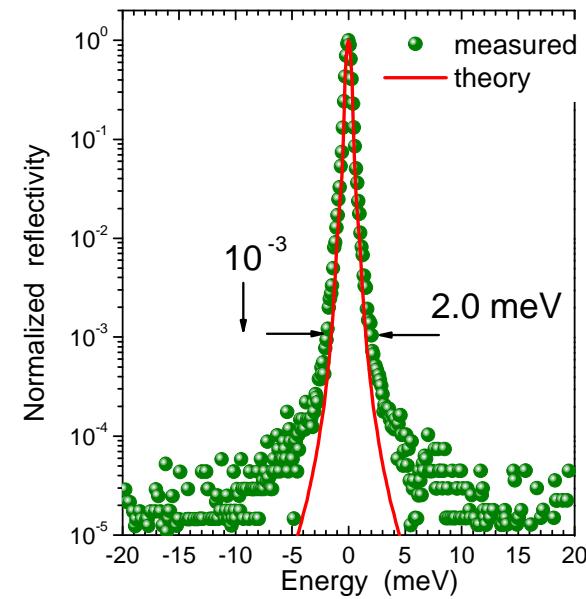
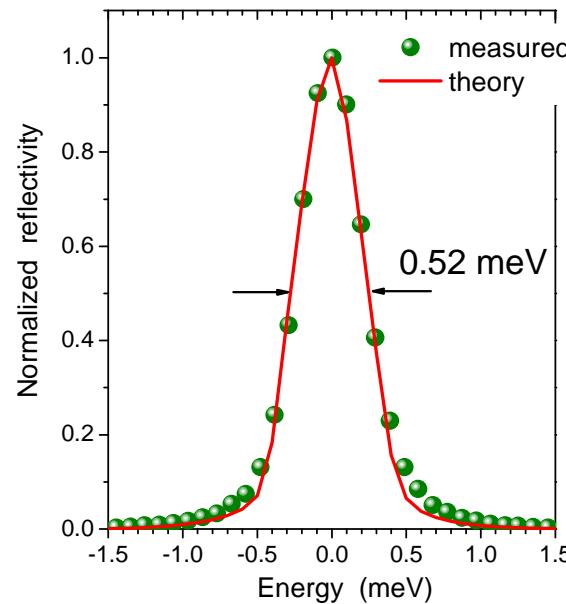
$\Delta E = \sim 0.5 \text{ meV}$, $R = \sim 40\%$, flux = $0.9 \times 10^{10} \text{ ph/s}$

$\Delta E = \sim 1.8 \text{ meV}$, $R = \sim 50\%$, flux = $4 \times 10^{10} \text{ ph/s}$

time to change monochromators: 25 sec

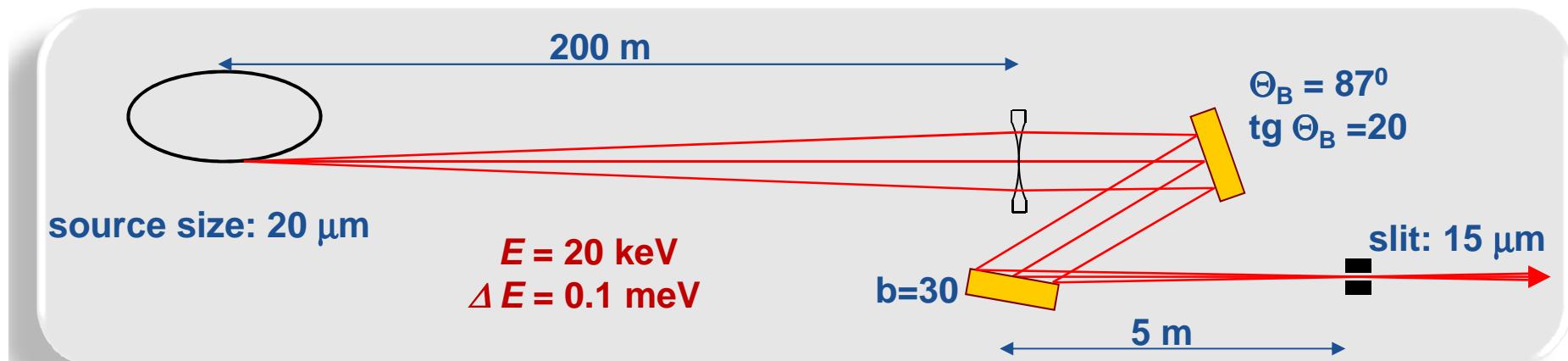
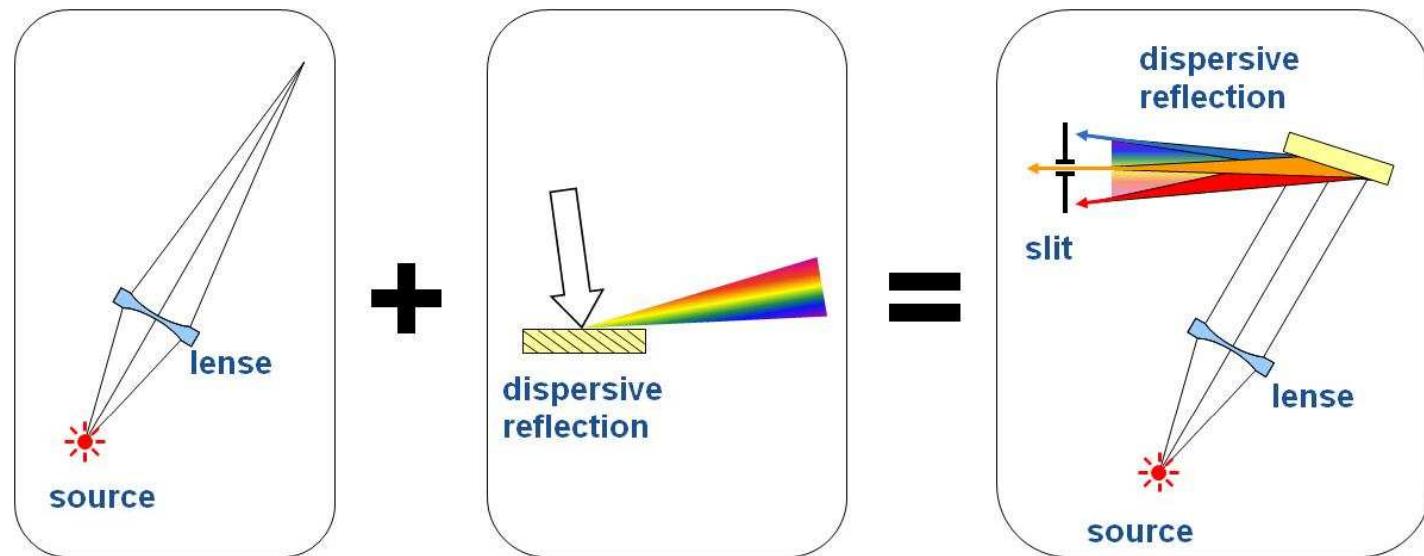


In-line monochromator with the bandwidth of ~0.5 meV



“Focusing” monochromator:

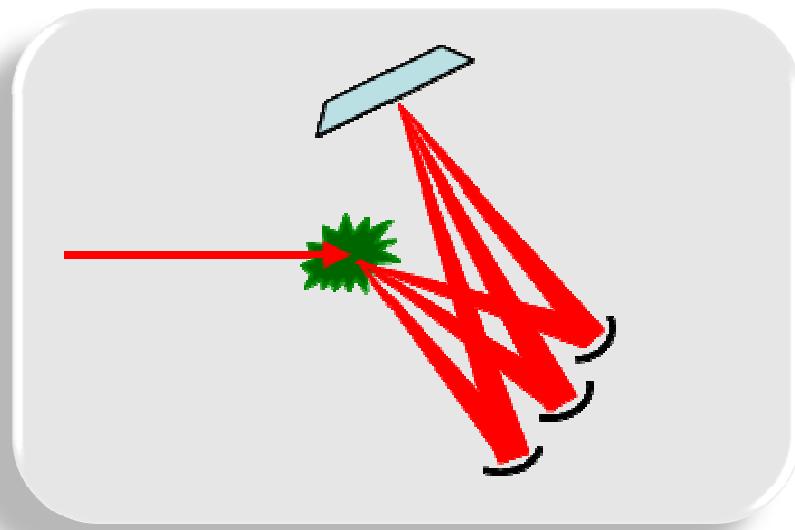
V.Kohn, ac, R.Rüffer, JSR **16** (2009) 635



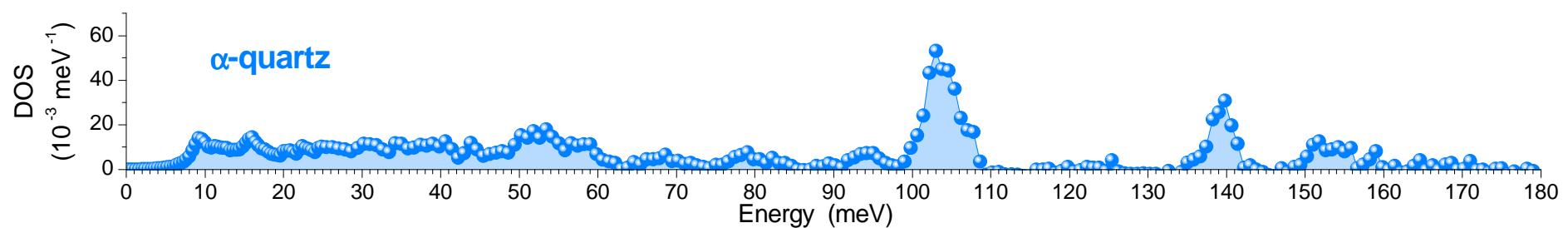
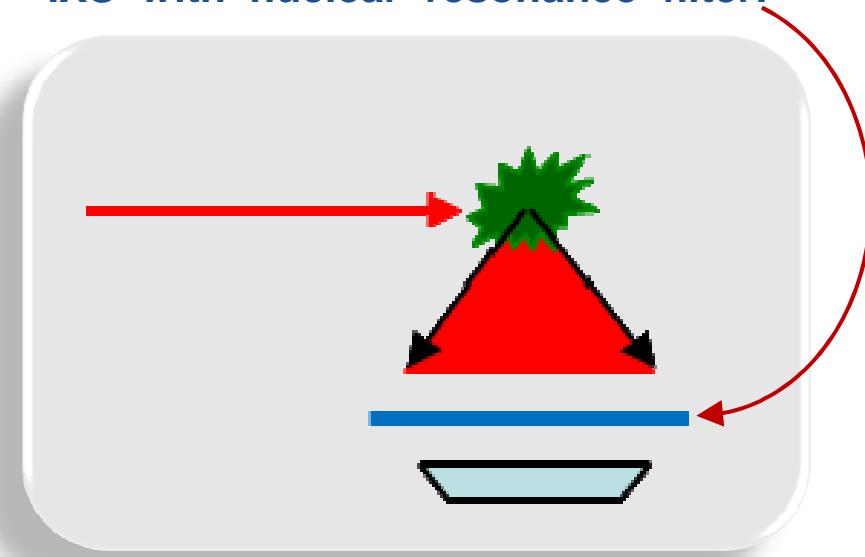
Nuclear Resonance analysis of IXS:

ac *et al*, PRL **76** (1996) 4258

IXS with crystal analyzers:



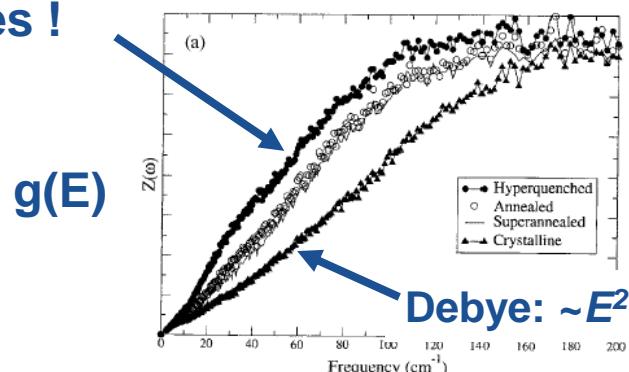
IXS with nuclear resonance filter:



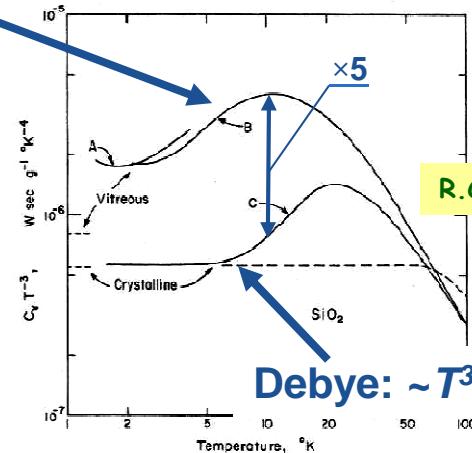
additional vibrational states?

At low temperatures,
heat capacity for glasses
is larger than for crystals

DOS $g(E)$:
additional
vibrational
states !



C.A. Angel et al.,
J.Phys.:Cond.Matt. 15, S1051, 2003

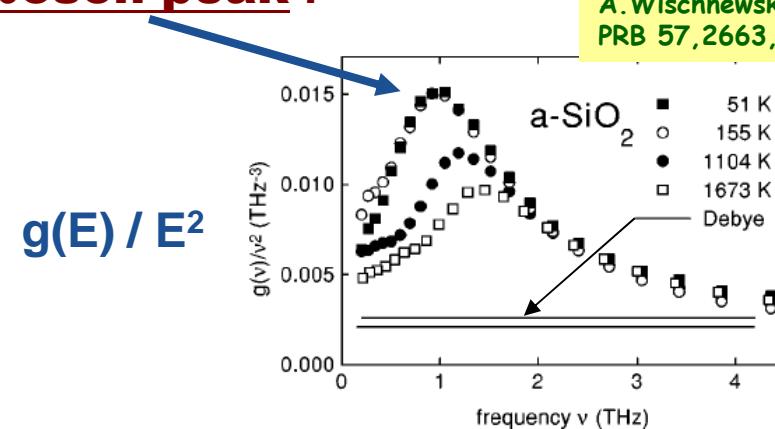


R.C.Zeller et al., PRB 4, 2029, 1971

Debye: $\sim T^3$

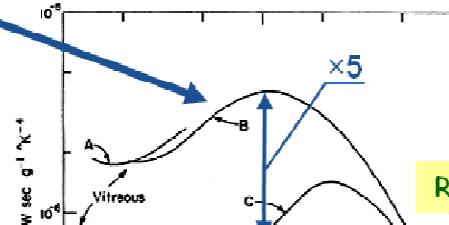
FIG. 4. Specific heat of vitreous SiO_2 and crystal quartz, plotted as $C_v T^{-3}$ vs T . A: I. R. Vitreosil (Ref. 29); B: vitreous silica (after Refs. 30–32); C: α -quartz

Reduced DOS $g(E)/E^2$:
the boson peak !



A.Wischnewski et al.,
PRB 57, 2663, 1998

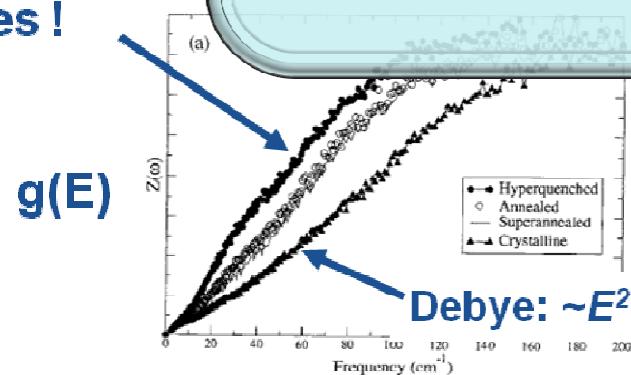
additional
vibrational states?



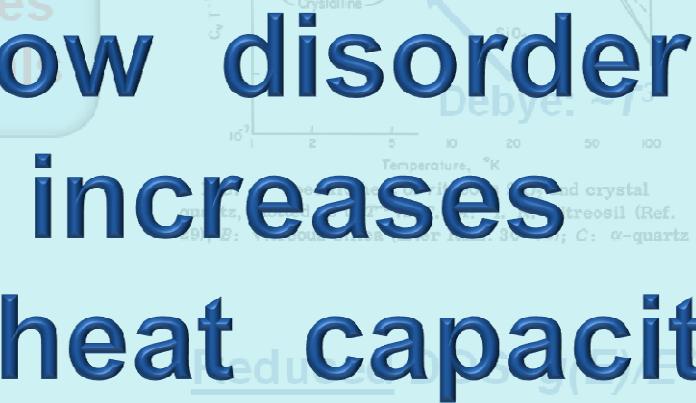
R.C.Zeller et al., PRB 4, 2029, 1971

At low temperatures
heat capacity for glasses
is larger than for crystals

DOS $g(E)$:
additional
vibrational
states !



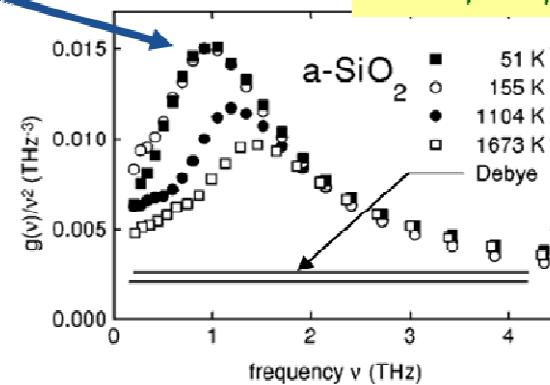
C.A.Angel et al.,
J.Phys.:Cond.Matt. 15, S1051, 2003



the boson peak!

Wischniewski et al.,
PRB 57, 2663, 1998

$g(E) / E^2$



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Source: NATURE Volume: 410 Issue: 6829 Pages: 663-667 DOI: 10.1038/35070517 Published:
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London University, Universität Bayreuth,
Max Planck Institut for the Science of Light, Aalborg University*

submitted to PRL

ambient glass



amorphous

densified glass



amorphous

 α -cristobalite

tetragonal

 α -quartz

trigonal

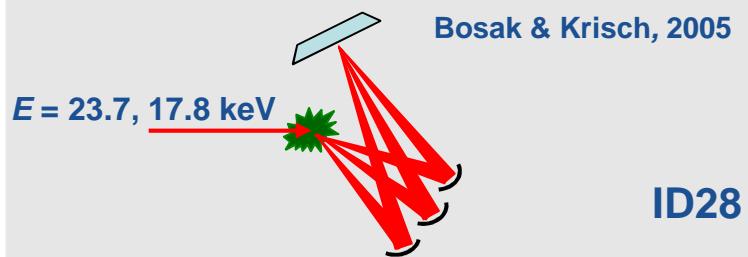
coesite



monoclinic

● Experimental method: Inelastic X-ray scattering (two methods)

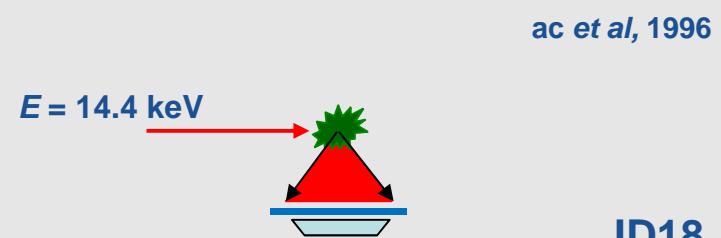
**with crystal analyzers:
(momentum-differentiated approach)**



ID28

$$\Delta E = 1.4 \text{ meV or } 3 \text{ meV}$$
$$\delta Q = 0.03 \text{ nm}^{-1}, \Delta Q = [1-7] \text{ nm}^{-1}$$

**with nuclear resonance analysis:
(momentum-integrated approach)**

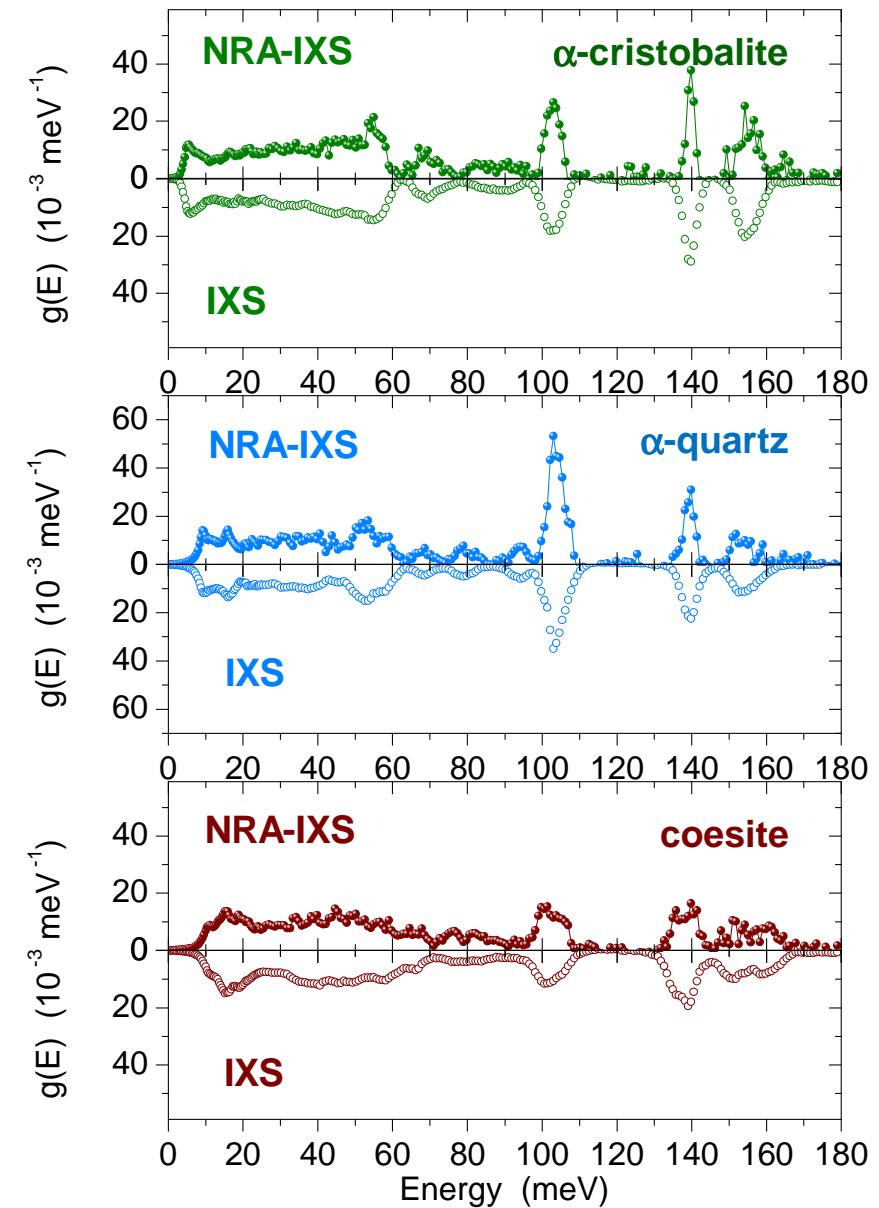
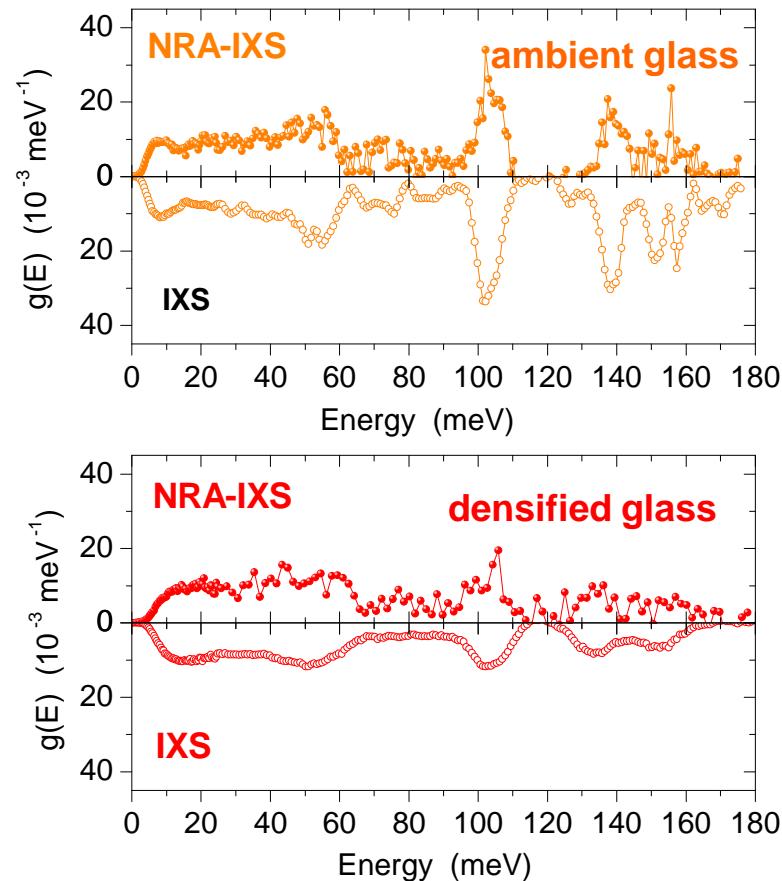


ID18

$$\Delta E = 0.7 \text{ meV}$$
$$\Delta Q = [3-14] \text{ nm}^{-1}$$

- Comparison of the data sets from two methods
- Comparison to ab initio calculations
- Comparison to heat capacity measurements
- Comparison to neutron data

- nuclear resonance analysis
- crystal analyzers

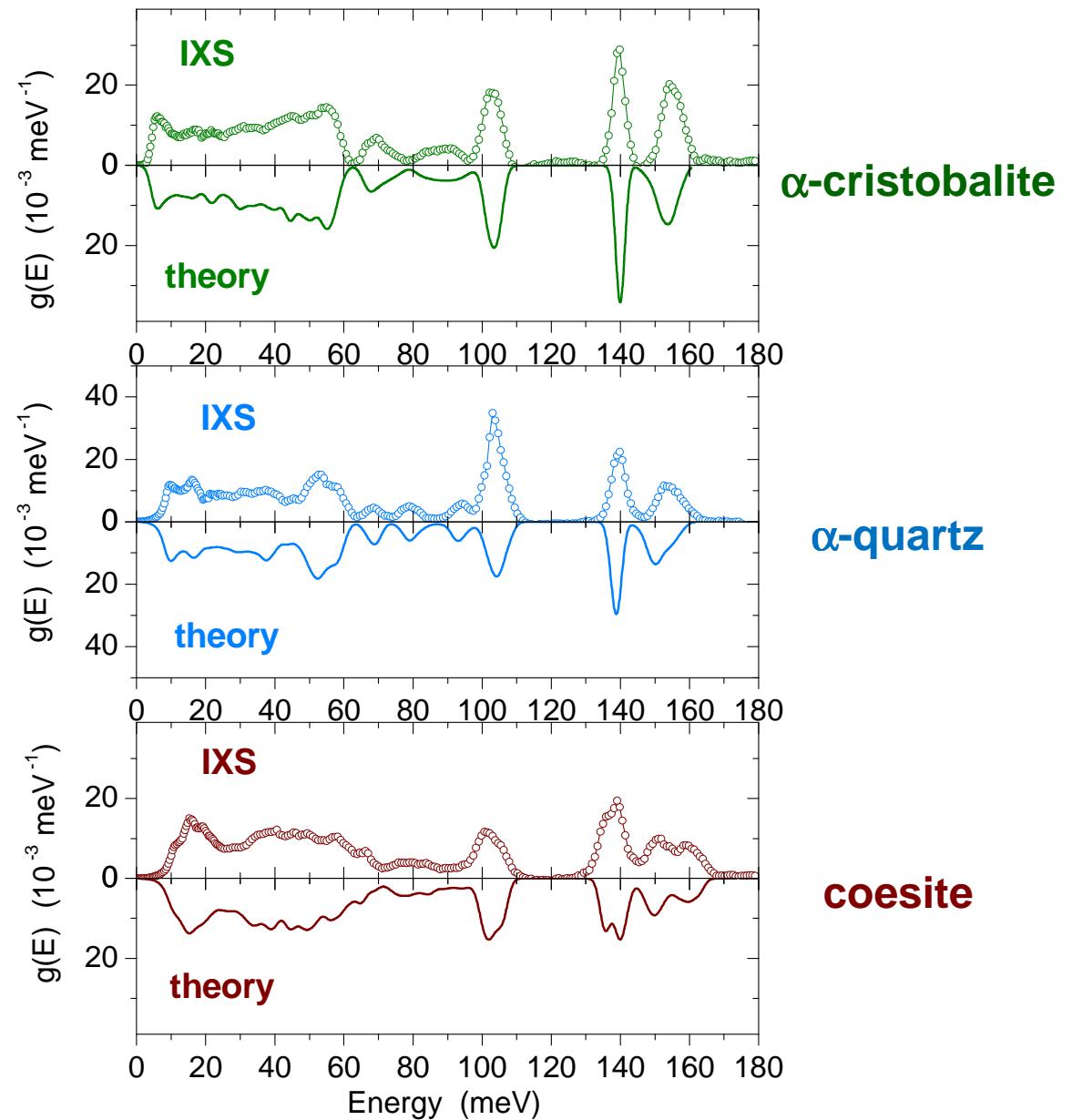


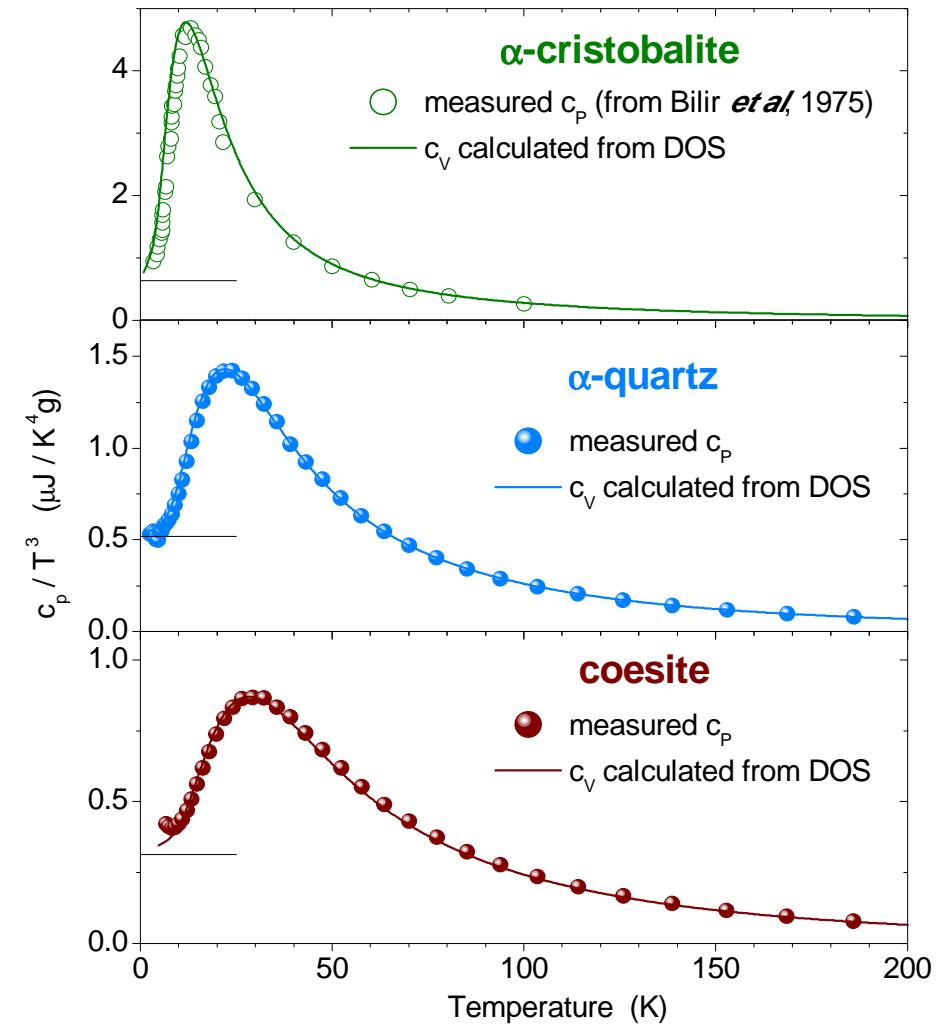
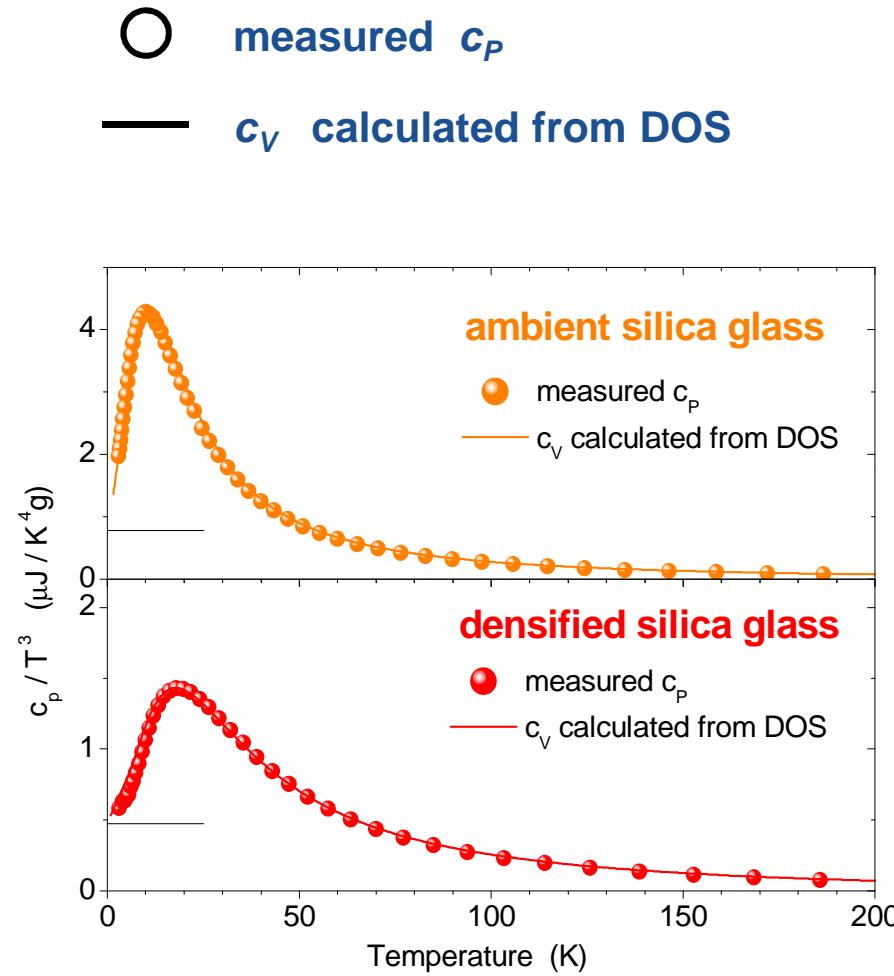


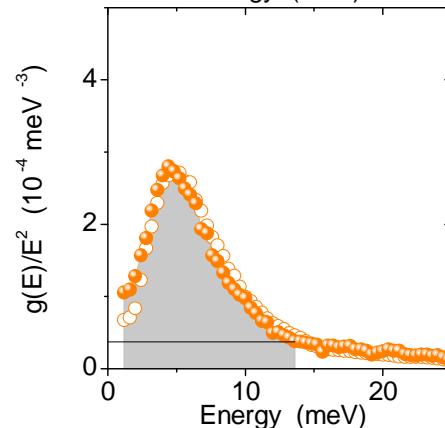
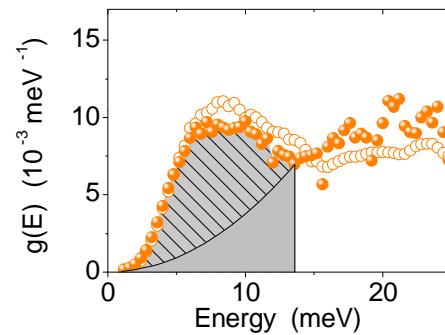
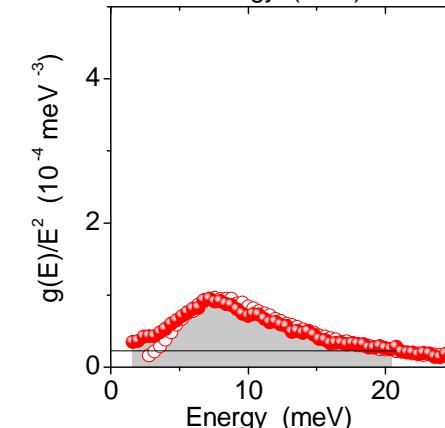
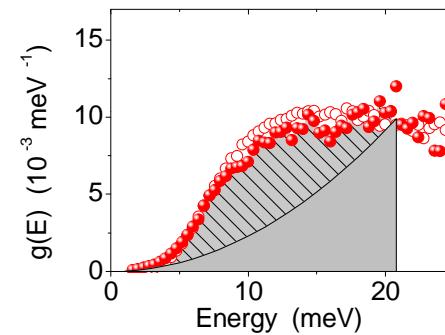
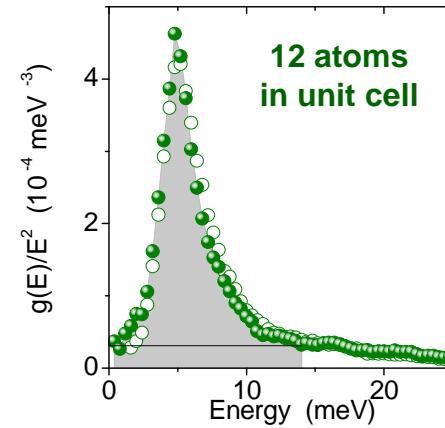
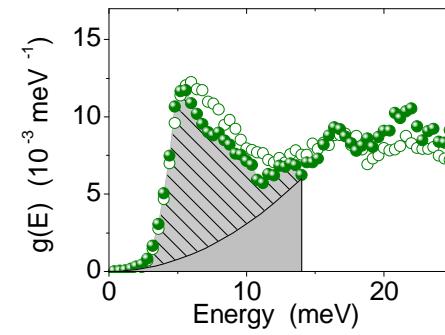
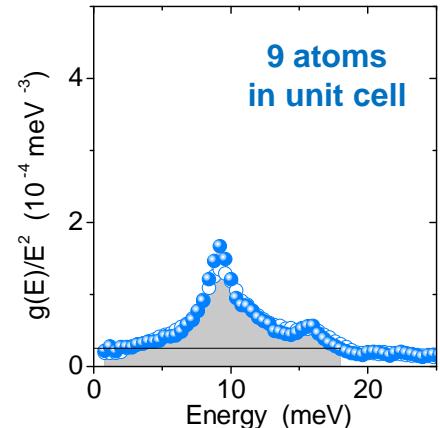
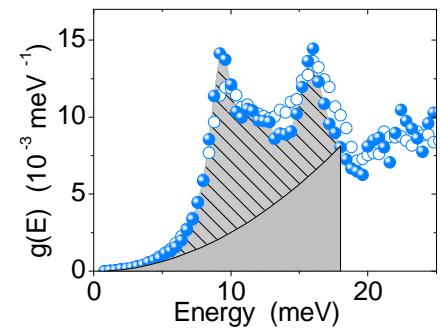
IXS, crystal analyzers

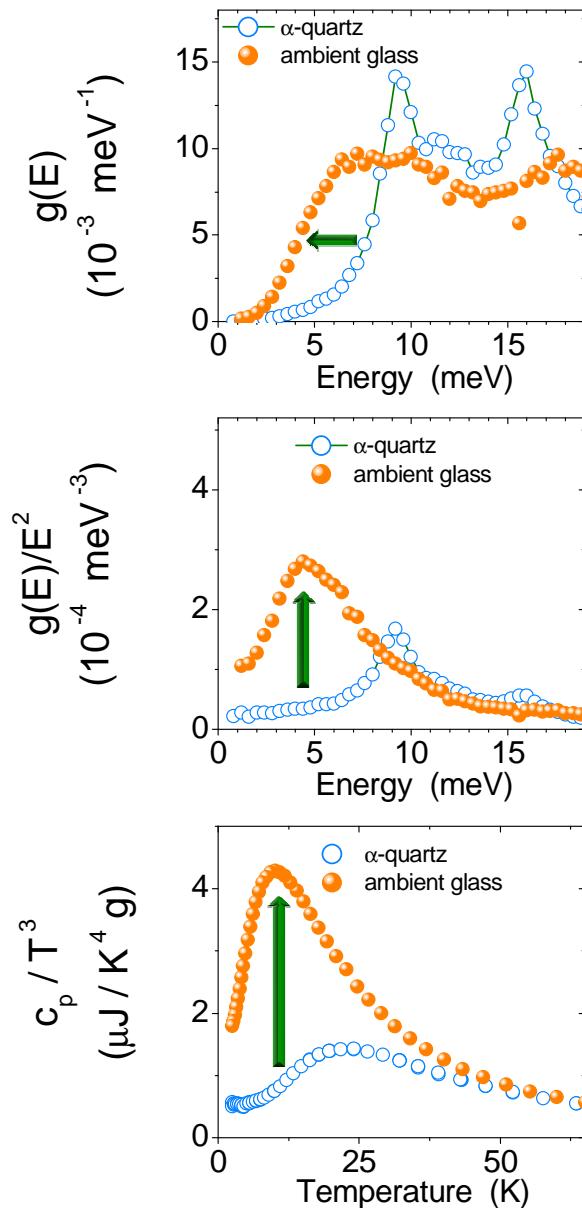


ab-initio calculations
convoluted
with instrumental function





ambient glass
excess states:
5.3(3)%
all states:
8.4(5)%

densified glass
excess states:
5.9(4)%
all states:
12.8(8)%

 α -cristobalite
excess states:
5.6(3)%
all states:
8.4(5)%

 α -quartz
excess states:
6.6(4)%
all states:
11.5(7)%




ambient glass



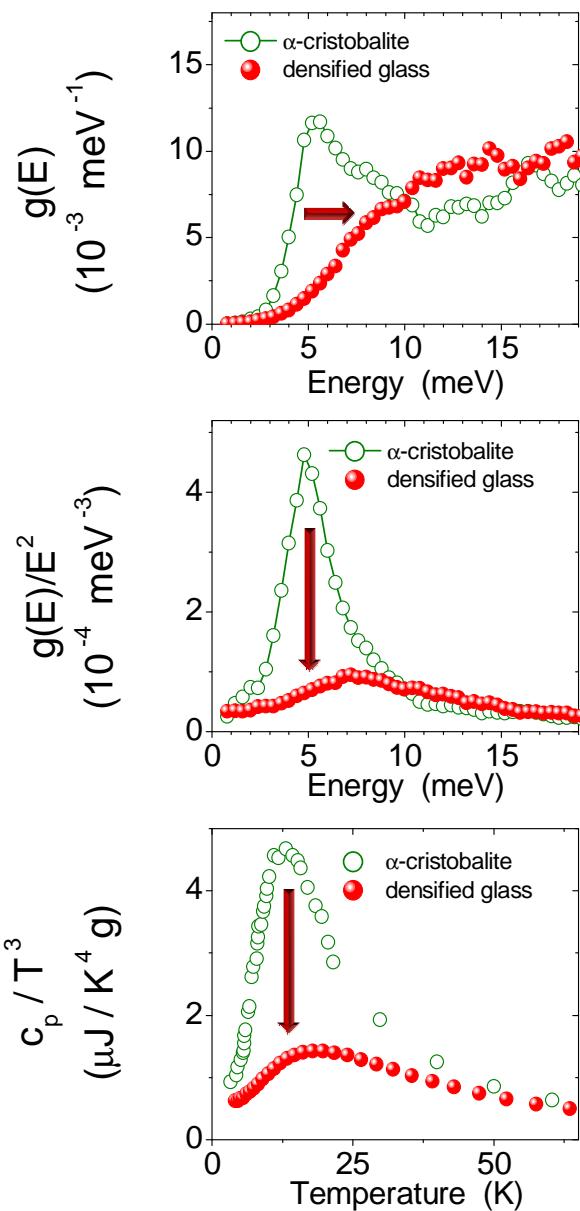
α -quartz



The role of disorder:

- shifts DOS to lower frequency,
- creates a peak in reduced DOS,
- provides excess of heat capacity.





densified glass



α -cristobalite

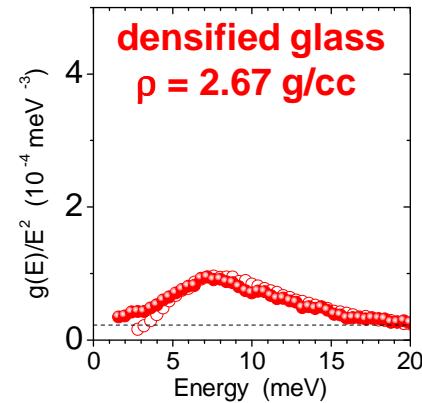
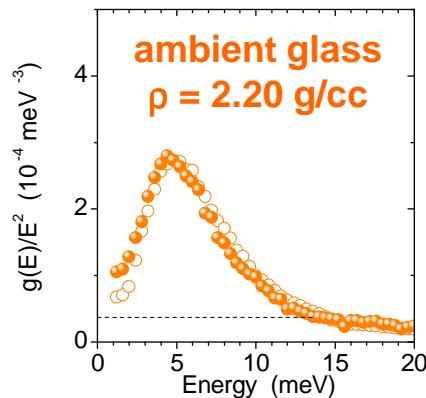


The role of disorder:

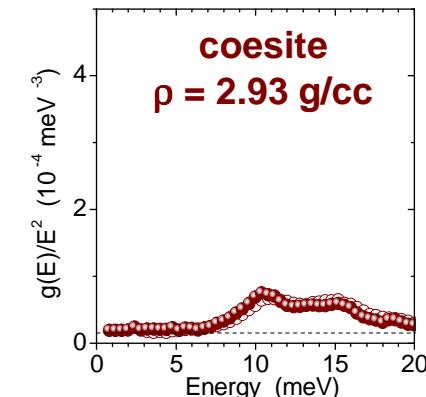
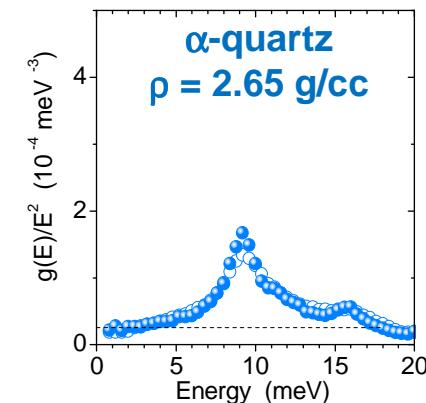
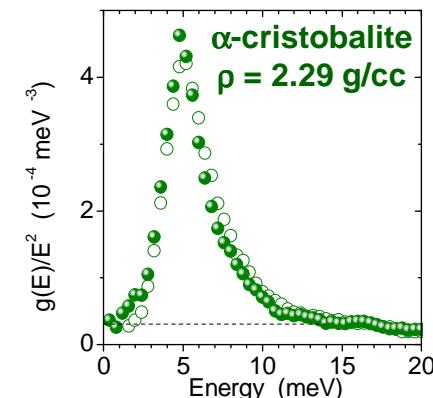
- shifts DOS to **higher frequency**,
- creates **flat reduced DOS**,
- provides **deficit of heat capacity**.



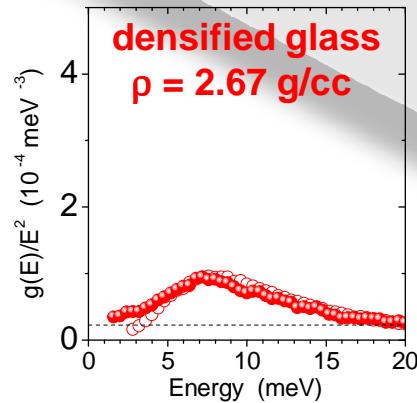
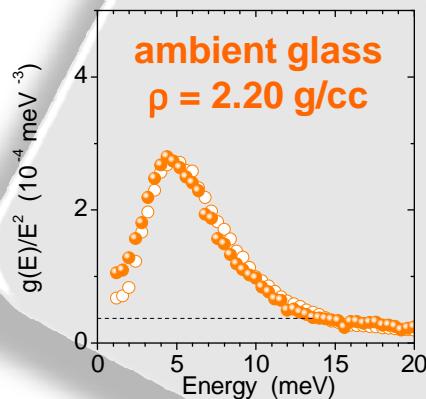
Where is the problem ?



DOS, reduced DOS,
and heat capacity
depend also on *density*:

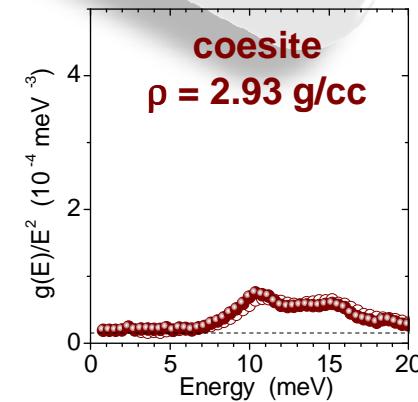
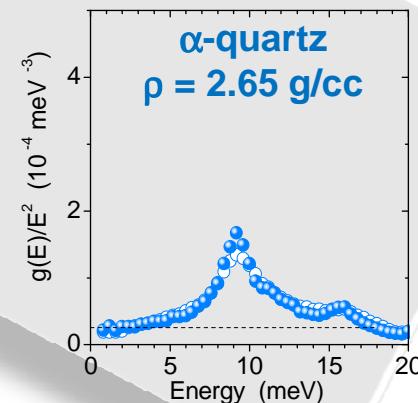
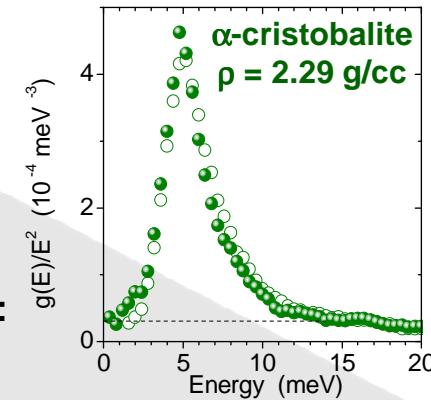


Where is the problem ?

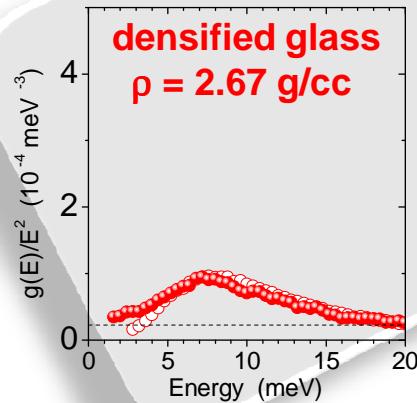
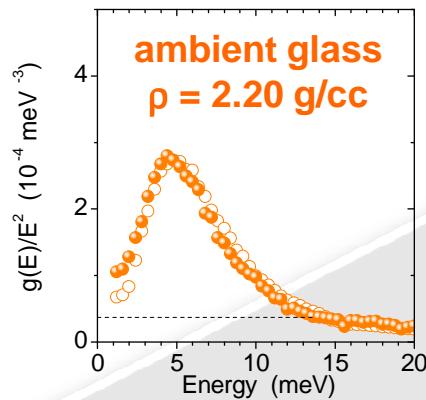


DOS, reduced DOS,
and heat capacity
depend also on *density*:

low-density glass
versus
high-density crystal:
disorder effect
plus
density effect

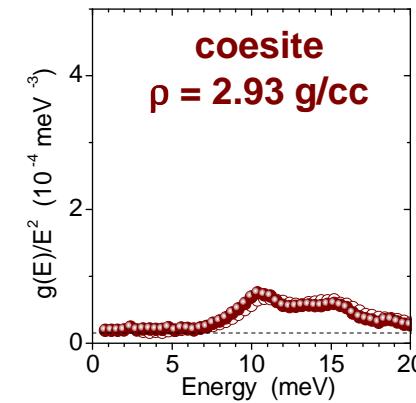
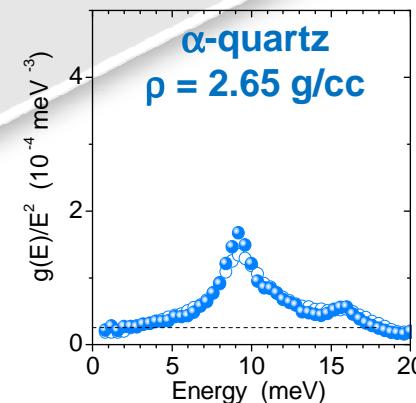
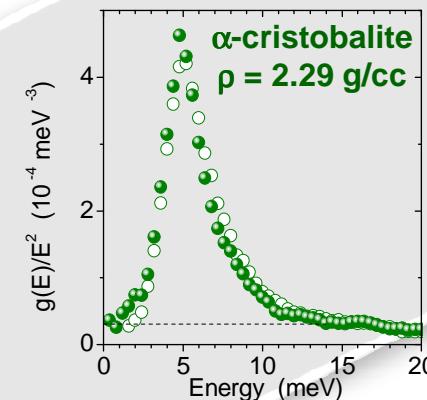


Where is the problem ?

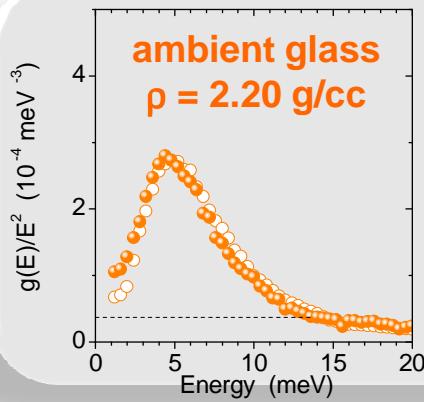


high-density glass
versus
low-density crystal:
disorder effect
minus
density effect

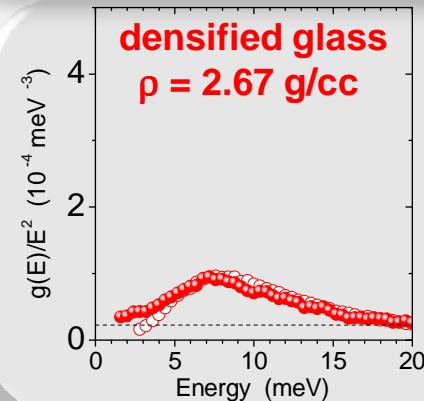
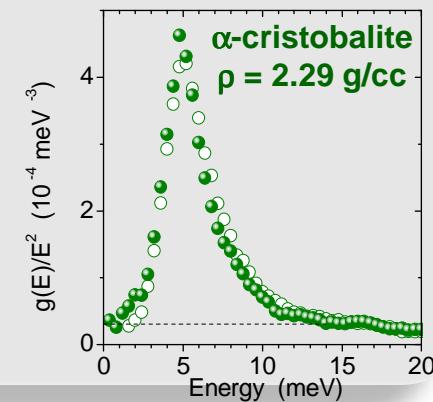
DOS, reduced DOS,
and heat capacity
depend also on density:



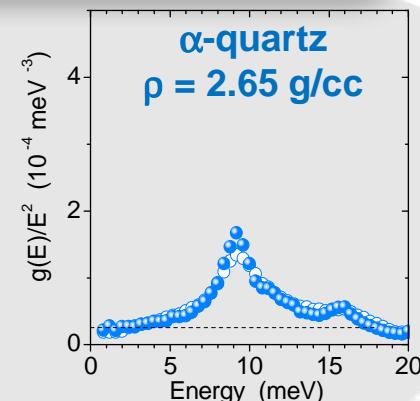
What should we compare?



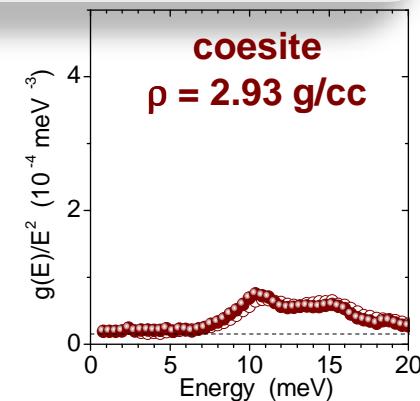
The low-density
glass and crystal:
only
the disorder effect

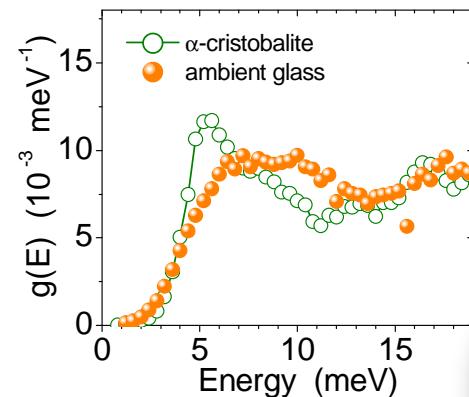


The high-density
glass and crystal:
only
the disorder effect

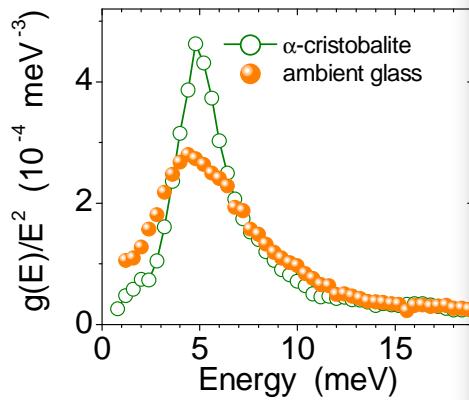


DOS, reduced DOS,
and heat capacity
depend also on density:



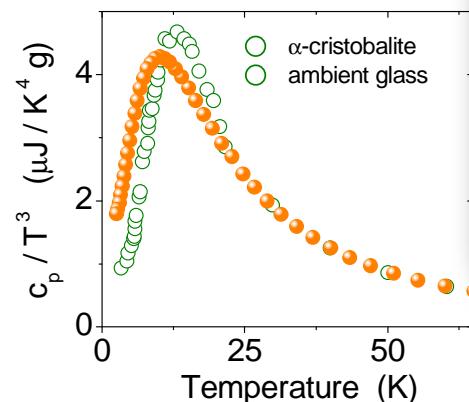


**ambient glass
 α -cristobalite**

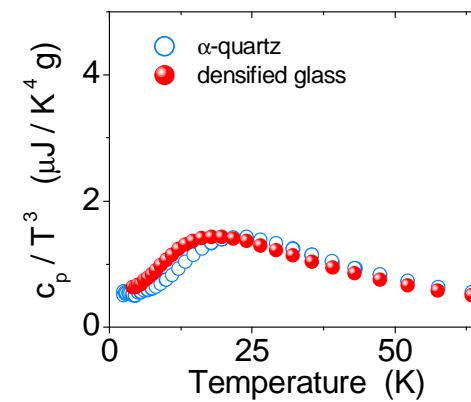
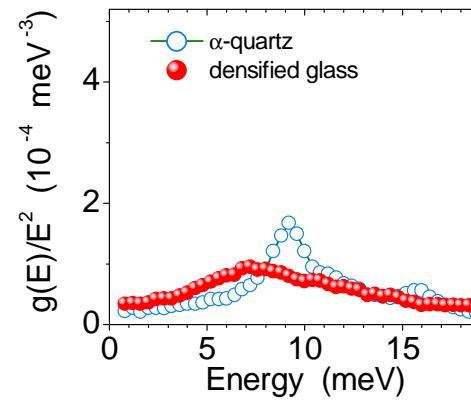
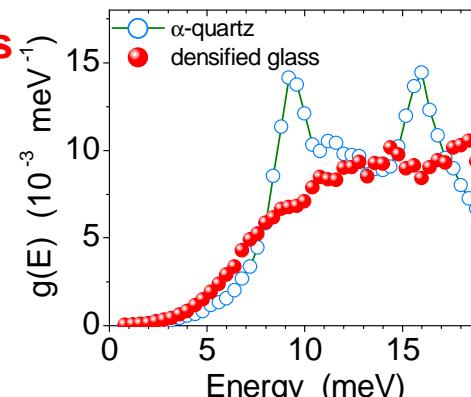


The role of disorder:

- does NOT make DOS softer,
- does NOT create peak in the reduced DOS,
the peak is already there,
- it smears out the DOS
and the reduced DOS,
- Disorder does NOT
change the heat capacity.



**densified glass
 α -quartz**



At low temperatures,
heat capacity for glasses
is larger than for crystals

How disorder increases the heat capacity?

R.C.Zeller et al., PRB 4, 2029, 1971

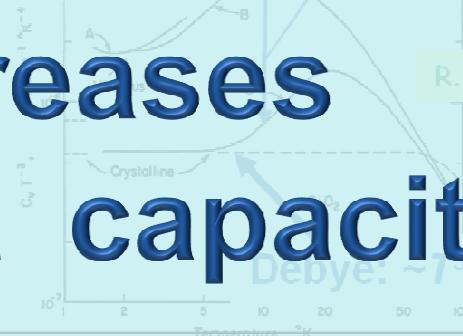
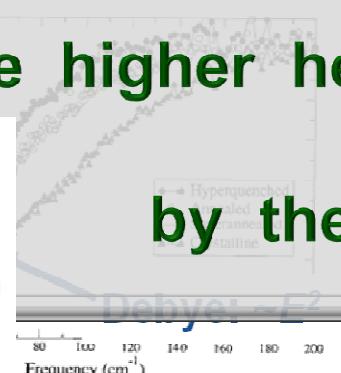


FIG. 1. Specific heat of vitreous SiO_2 and crystalline quartz, plotted as $C_v T^3$ vs T . A: I. R. Vitreosil (Ref. 29); B: vitreous silica (after Refs. 30–32); C: α -quartz

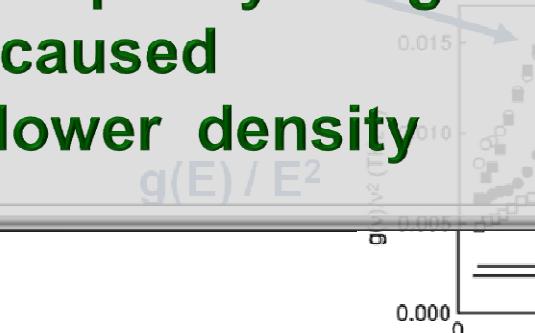
DOS $g(E)$:
additional
vibrational
states !

It does not do it:
the higher heat capacity of glasses
is caused
by their lower density

A. Wischnewski et al.,
PRB 57, 2663, 1998



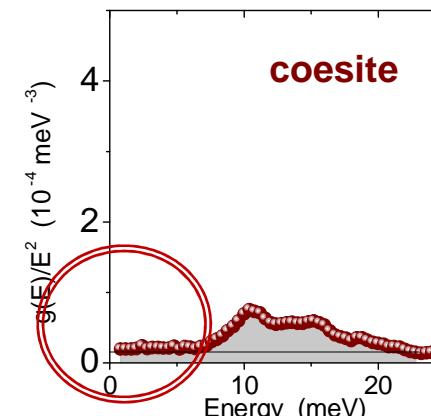
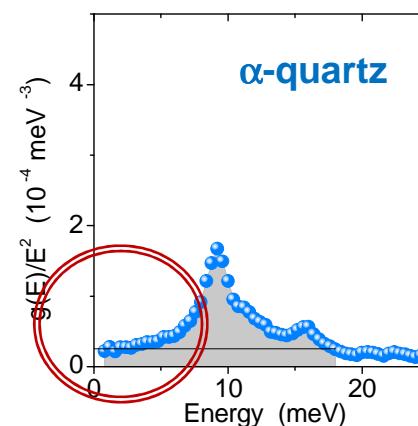
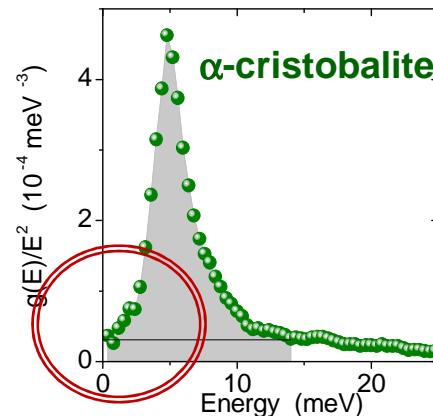
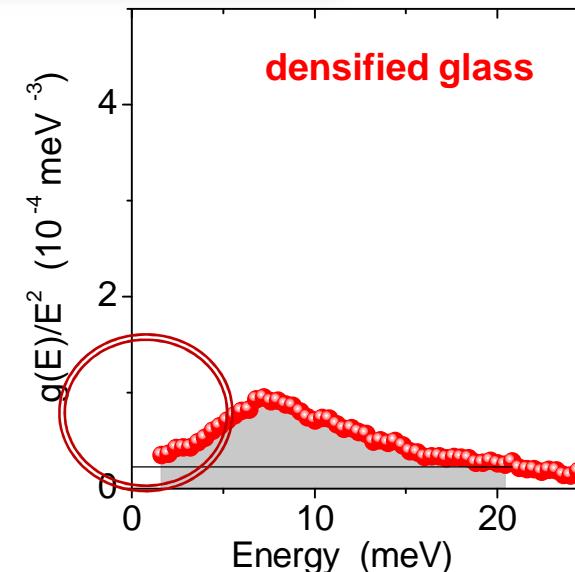
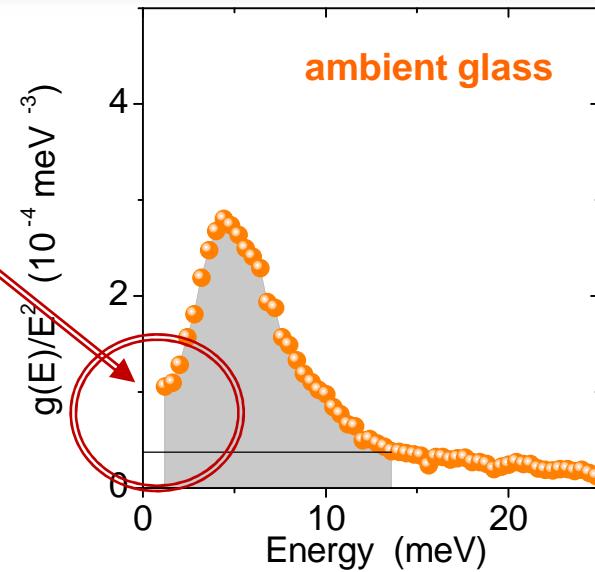
Engel et al.,
Phys.: Cond. Matt. 15, S1051, 2003



Are there any unusual soft modes at lower energy?

What is that?

- onset of instability?
- anharmonic effects?
- Does reduced DOS go to the Debye level?



What is the origin of quasi-elastic scattering?

B.Frick *et al*, PRB **47** (1993) 14795

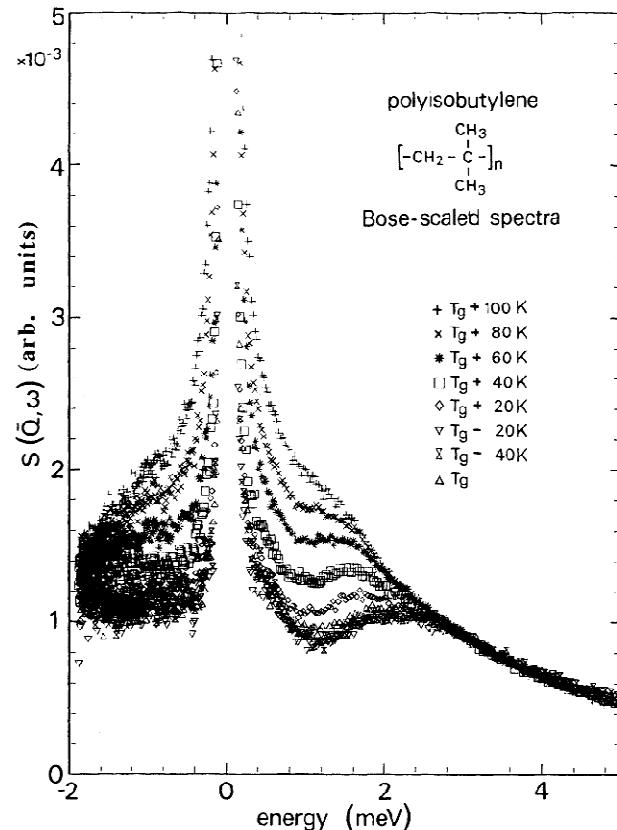


FIG. 6. Dynamic structure factor $S(Q, \omega)$ of PIB corrected for Debye-Waller factor and Bose factor as measured on IN6, ILL. The measured curves are presented for $T=200\text{ K}$, the glass-transition temperature of PIB, as a reference temperature.

relaxation?
vibrations?

A.P.Sokolov *et al*, PRB **57** (1997) 5042

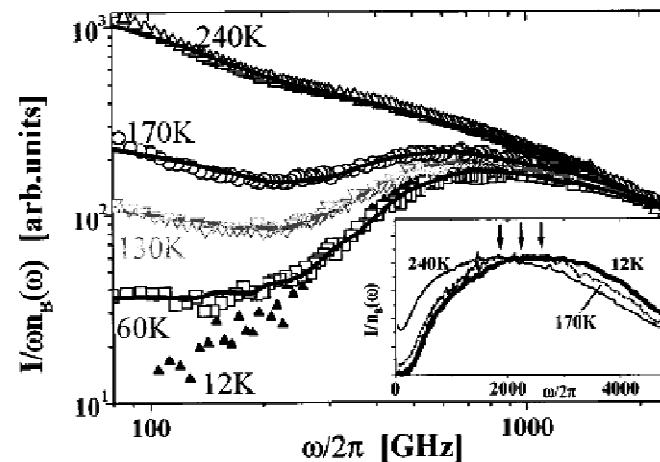
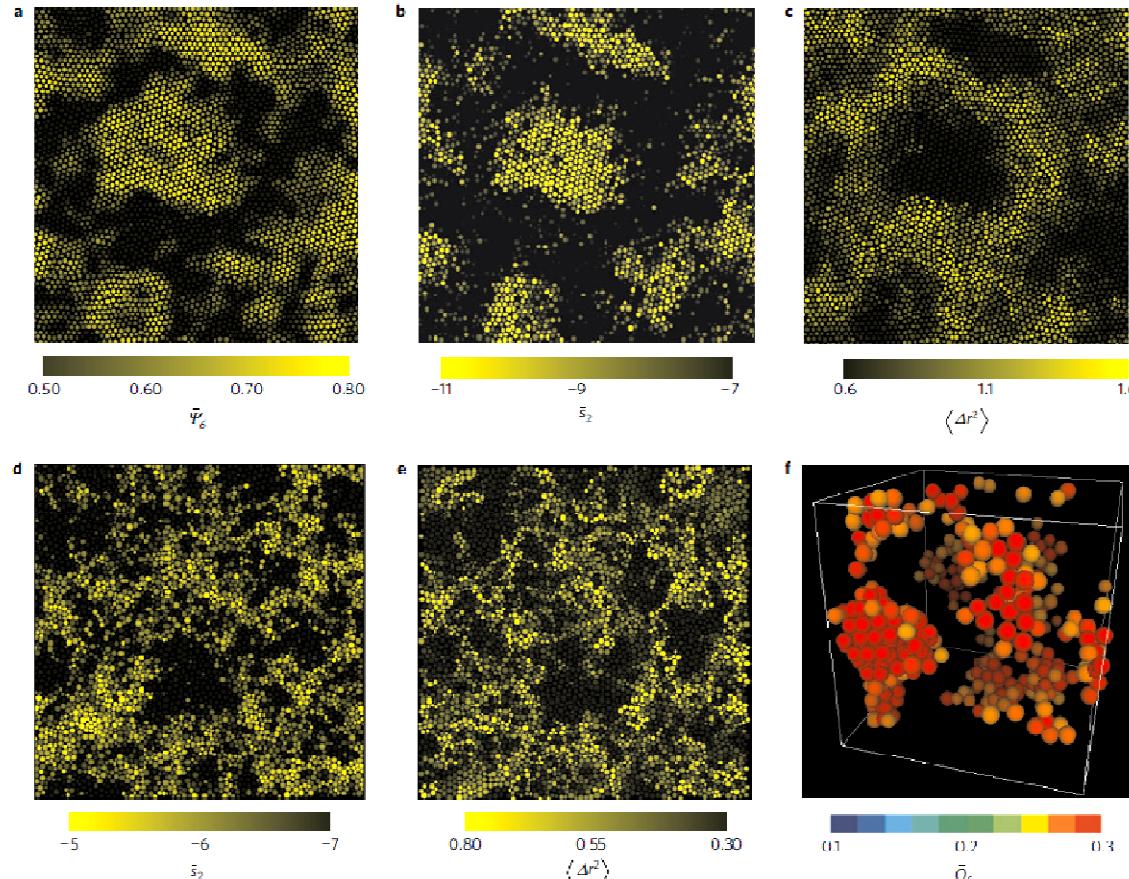


FIG. 3. The Raman spectra of PB at different temperatures (symbols) and results of the fit (lines). The inset shows the Raman susceptibility spectra and the arrows show the shift of Ω_{end} with temperature.

Can we directly observe dynamical heterogeneities?



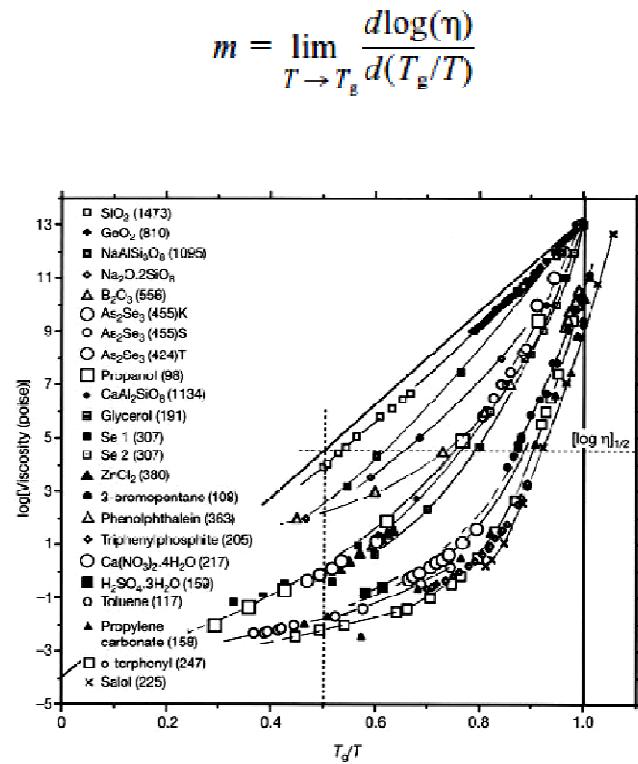
H.Tanaka *et al*, Nmat **9** (2010) 324

can we
see this?

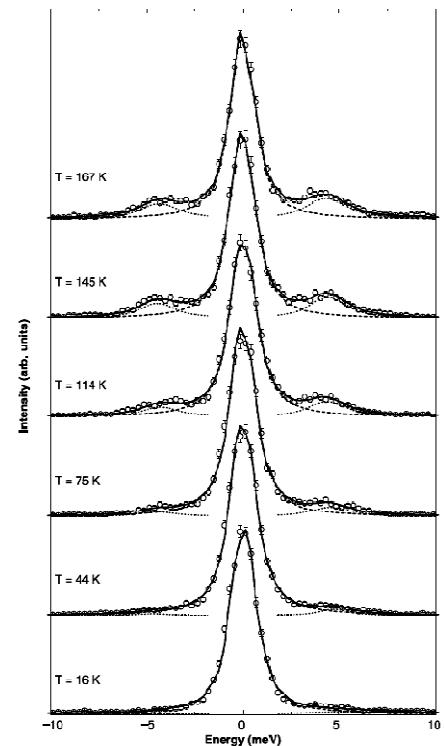
Figure 1 | Structural order in glass-forming liquids. **a**, Spatial distribution of $\bar{\psi}_6$ for 2DPC with $\Delta_{2DPC} = 9\%$ at $\phi = 0.740$. See Supplementary Video S1 for dynamical fluctuations of $\bar{\psi}_6$. **b**, Spatial distribution of \bar{s}_2 for the same system as in a. **c**, Spatial distribution of the mean-square displacement $\langle \Delta r^2 \rangle$ over $t = 10\tau_d$ for the same system as in a. **d**, Spatial distribution of the local structural entropy \bar{s}_2 , which is averaged over $10\tau_d$. **e**, Spatial distribution of the mean square displacement $\langle \Delta r^2 \rangle$ over $10\tau_d$, $\langle \Delta r^2 \rangle(10\tau_d)$. **f**, Spatial distribution of \bar{Q}_6 for 3DPC with $\Delta_{3DPC} = 6\%$ at $\phi = 0.557$. Here, less ordered particles are not shown for clarity. See also Supplementary Fig. S1a,b for 2DGL and 2DSL, respectively.

Why “slow” viscosity correlates with fast atomic vibrations?

L.-M.Martivez and C.A.Angell, Nature **410** (2001) 663

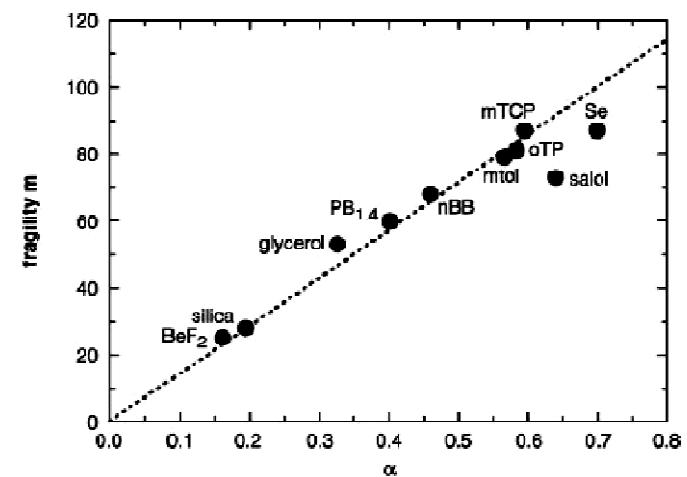


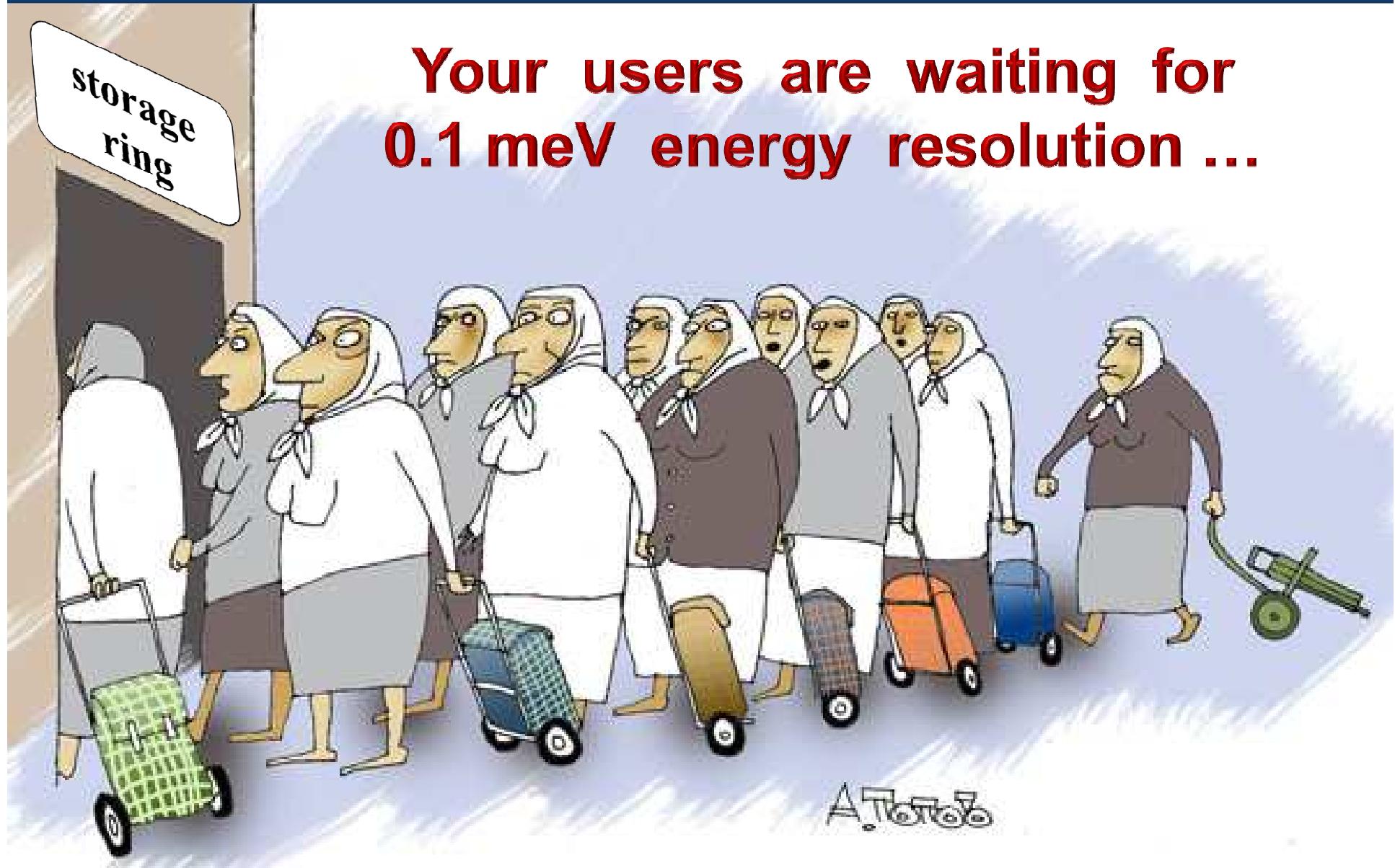
T.Scopigno *et al*, Science **302** (2003) 849



$$f(Q, T) = \lim_{t \rightarrow \infty} \Phi_Q(t) =$$

$$\lim_{t \rightarrow \infty} \frac{S_{\text{IS}}(Q) + F_{\text{inel}}(Q, t)}{S_{\text{IS}}(Q) + S_{\text{inel}}(Q)} = \frac{1}{1 + S_{\text{inel}}(Q)/S_{\text{IS}}(Q)}$$





Thank you
for
your attention

